

Ba_{0.7}Sr_{0.3}TiO₃ Ceramic Fibers Prepared via Precursors Linear Self-Assembly Nonhydrolytic Sol-Gel Method

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Abstract:

Ba_{0.7}Sr_{0.3}TiO₃ fibers were prepared via precursor self-assembly non hydrolytic sol-gel (NHSG) method, using barium acetate, titanium tetrachloride and strontium acetate as barium source, titanium source and strontium source, anhydrous ethanol as oxygen donor and anhydrous ethylene glycol as solvent. It is discovered that NHSG method is beneficial for Ba–O–Ti and Sr–O–Ti bonds formation via heterogeneous condensation. Bimolecular association structure of chlorotitanium ethoxide generated from reaction of titanium tetrachloride and ethanol, which is beneficial for precursors' self-linear assembling. Linear colloids with excellent spinnability form in this process. The novel Ba_{0.7}Sr_{0.3}TiO₃ ceramic fibers will be widely used in electronic flexible products.

Keywords: Ceramics, Ba_{0.7}Sr_{0.3}TiO₃, Fibers, Dielectrics, Piezoelectric.

I. INTRODUCTION

As the most important electronic components, capacitors account for a large part of the electronic components [1]. Output value of global capacitor components has broken through billions of dollars (US). With the develop rapidly of the information industry, the requirements for the performance of capacitors are getting higher and higher [2].

For the requirements to multifunction, miniaturization and flexibility of electronic products, components of capacitor should be flexibility, large capacity and miniaturization [3]. Ba_{0.7}Sr_{0.3}TiO₃ ceramic fiber has broad application prospects in embedded packaging components and high-power

capacitors due to high energy storage density, high flexibility and dielectric constant [4-5]. Recent contributions have devoted to relative topics [6-8]. Pan [6] fabricated BST ($\text{Ba}_x\text{Sr}_{1-x}\text{TiO}_3$)/ PVDF (polyvinylidene fluoride) fiber composites via near-field electrospinning (NFES) method, in which BST ceramic powders were mixed with PVDF to prepare piezoelectric fiber composites. Vigneshwaran [7] and He [8] investigated the effect of calcined time and temperature on structure and properties of $\text{Ba}_x\text{Sr}_{(1-x)}\text{TiO}_3$ ceramics. Our team has developed a novel precursors linear self-assembly nonhydrolic sol-gel method [9]. In this work, the method is utilized to prepare novel $\text{Ba}_{0.7}\text{Sr}_{0.3}\text{TiO}_3$ ceramic fibers.

II. MATERIALS AND METHODS

2.1. Samples Preparation

Titanium tetrachloride (AR, TiCl_4 , Shanghai), anhydrous barium acetate (AR, $\text{Ba}(\text{OOCCH}_3)_2$, Shanghai), anhydrous strontium acetate (AR, $\text{Sr}(\text{OOCCH}_3)_2$, Shanghai), glycol (AR, $(\text{CH}_2\text{OH})_2$, Shanghai) and absolute ethanol (AR, EtOH, Shanghai) were selected as the raw material.

In accordance with the nominal composition of $\text{Ba}_{0.7}\text{Sr}_{0.3}\text{TiO}_3$ (the relative molar ratio of $\text{Ba}(\text{OOCCH}_3)_2 : \text{Sr}(\text{OOCCH}_3)_2 : \text{TiCl}_4 = 0.7 : 0.3 : 1$), add 4.1mL TiCl_4 slowly (about 10 mL/min) through a pipette in glove box into a 5.5mL anhydrous EtOH Erlenmeyer flask (100mL). After stirring for 6h, a transparent bright yellow TiCl_4 -EtOH solution forms. After that, 17mL glycol is added to solution and stirring for 30 min. Then, 6.670g $\text{Ba}(\text{OOCCH}_3)_2$ and 2.403g $\text{Sr}(\text{OOCCH}_3)_2$ are added into the solution. And then, the solution is fluxed and agitated at 80°C for 2h and further aged at 60°C without stirring for 48h, obtaining spinnable sol with the viscosity of about 50Pa·S. The spinnable sol is fiber-formed by drawing with a glass rod at the speed of 0.5m/s. After fiber-forming, $\text{Ba}_{0.7}\text{Sr}_{0.3}\text{TiO}_3$ wet gel fibers are obtained, and they are dried at 80°C for 4h. Xerogels fibers were finally calcined at 700°C for 0.5 h to form the final fibers.

2.2 Characterization

Under the conditions of 40kV and 30mA, the phase of the sample was determined by X-ray diffractometer (XRD, Bruker) radiation using $\text{CuK}\alpha$ ($\lambda=1.5405\text{\AA}$). Samples diffractograms were recorded in the 2θ range of 10 - 70° in steps of 0.02° with the count time of 0.2s (5s for XRD structural refinement). Morphology of sol is characterized by transmission electron microscopy (TEM, JEOL). The microstructure of fibers is observed by field-emission scanning electronmicroscopy (FE-SEM, JEOL). The thermolysis process of xerogels is tested by thermogravimetric and differential thermal analysis analysis (DTA-TG, NETZSCH) (Heating rate: $10^\circ\text{C}/\text{min}$, test temperature range: room temperature- 900°C , test atmosphere: air atmosphere). The bonds of the samples are tested in the range of 400 - 4000 cm^{-1} by Fourier transform infrared spectroscopy (FT-IR, Thermo).

III. RESULTS AND DISCUSSION

Fig. 1(a) displays the DTA-TG patterns of fibers xerogel. Fig. 1(b) presents corresponding XRD patterns.

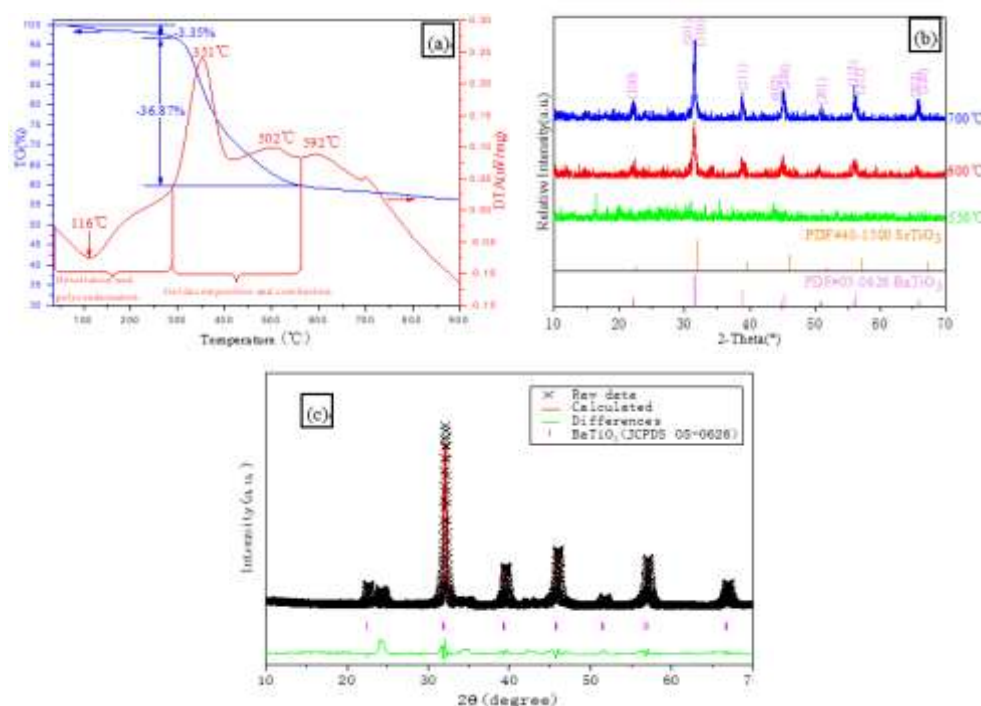


Fig 1: DTA-TG curves of xerogel (a), XRD patterns of samples prepared at different temperatures (b) and refinement of XRD pattern (c)

DTA curve of Ba_{0.7}Sr_{0.3}TiO₃ ceramic xerogel presents an obvious endothermic peak located at 116°C, which corresponds to 3.35wt.% weight loss from the corresponding temperature range (room temperature~270°C) on the TG curve. This is due to the further solvation and condensation of xerogels. The second exothermic peak cauterized at 351°C, which is due to the decomposition of the xerogel. The third exothermic peak locates at 502°C, and it is caused by the combustion of the xerogel. The fourth exothermic peak is concentrated at 592°C. Combined with the samples calcined at 500°C and 600°C without any obvious weight loss and phase transformation, and the exothermic peak at 592°C is confirmed to the crystallization of BaTiO₃. The sample crystallinity calcined at 700°C is higher than that of sample at 600°C. Compared with the standard diffraction card of BaTiO₃ (JCPDS Card No. 05-0626), all diffraction peaks move slightly to a higher angle. In addition, rietveld refinement is used to analyze the cell parameters of barium strontium titanate and Fig.1(c) shows the refinement results. The cell

parameters before and after refinement are $a=b=c=3.985$ (Vol = 63.283) and 3.961 (Vol = 62.159), respectively. These results indicate Sr^{2+} has been doped into the crystal lattice of barium titanate.

The TEM of the sol is shown as Fig. 2 (a) and (b). As can be seen from Fig. 2(a), there are clear needle-like or rod-like colloids. The diameters of the needle-like or rod-like colloids are about $0.1 - 0.25\mu\text{m}$, and the aspect ratios of $\text{Ba}_{0.7}\text{Sr}_{0.3}\text{TiO}_3$ ceramic fibers are larger than 10. The higher magnification shown as Fig. 2(b) shows the linear sol particle is in solid rod state. These results confirm that the precursors can linearly self-assemble to needle-like or rod-like colloids which are widely regarded as the premise of fiber forming by sol-gel method. Although the surface of colloids is not very smooth and there are also granular colloids with blurred outline. Fig. 2(c) (d) is FE-SEM images of $\text{Ba}_{0.7}\text{Sr}_{0.3}\text{TiO}_3$ ceramic fibers, it can be seen that it is a typical SEM micrograph of the fiber. The mean diameter of $\text{Ba}_{0.7}\text{Sr}_{0.3}\text{TiO}_3$ ceramic fibers is determined as about $2-2.5\mu\text{m}$. The aspect ratios of $\text{Ba}_{0.7}\text{Sr}_{0.3}\text{TiO}_3$ ceramic fibers are larger than 10. Although there are obvious a few spotted defects on both the surface and the cross-section of the fibers, the fibers are overall smooth in surface and in solid state in cross-section.

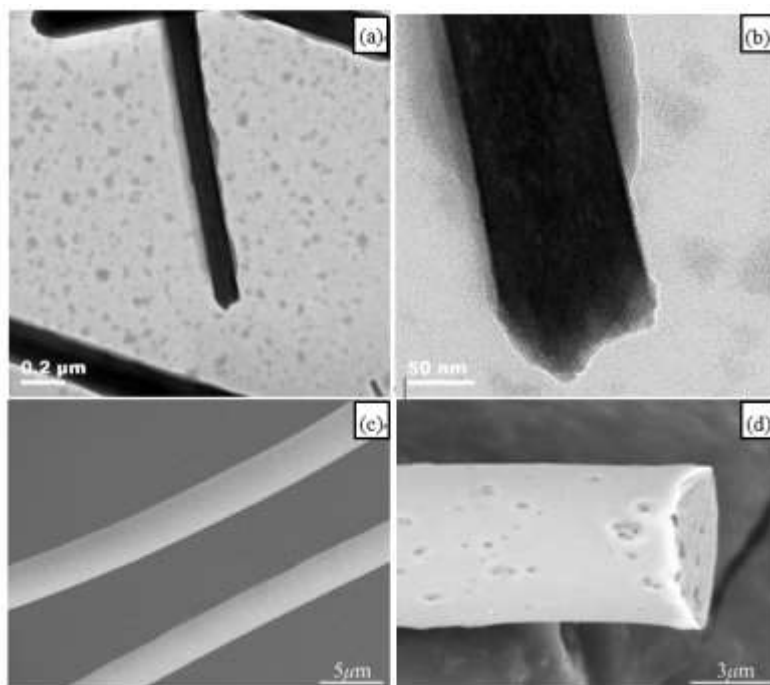


Fig 2: TEM graphs of sol (a) (b) and FE-SEM graphs of fibers (c) (d)

Fig.3 shows the EDS-mapping of fibers. From Fig.3 can be seen, the fibers show considerable homogeneity and highly consistent in composition with the nominal composition.

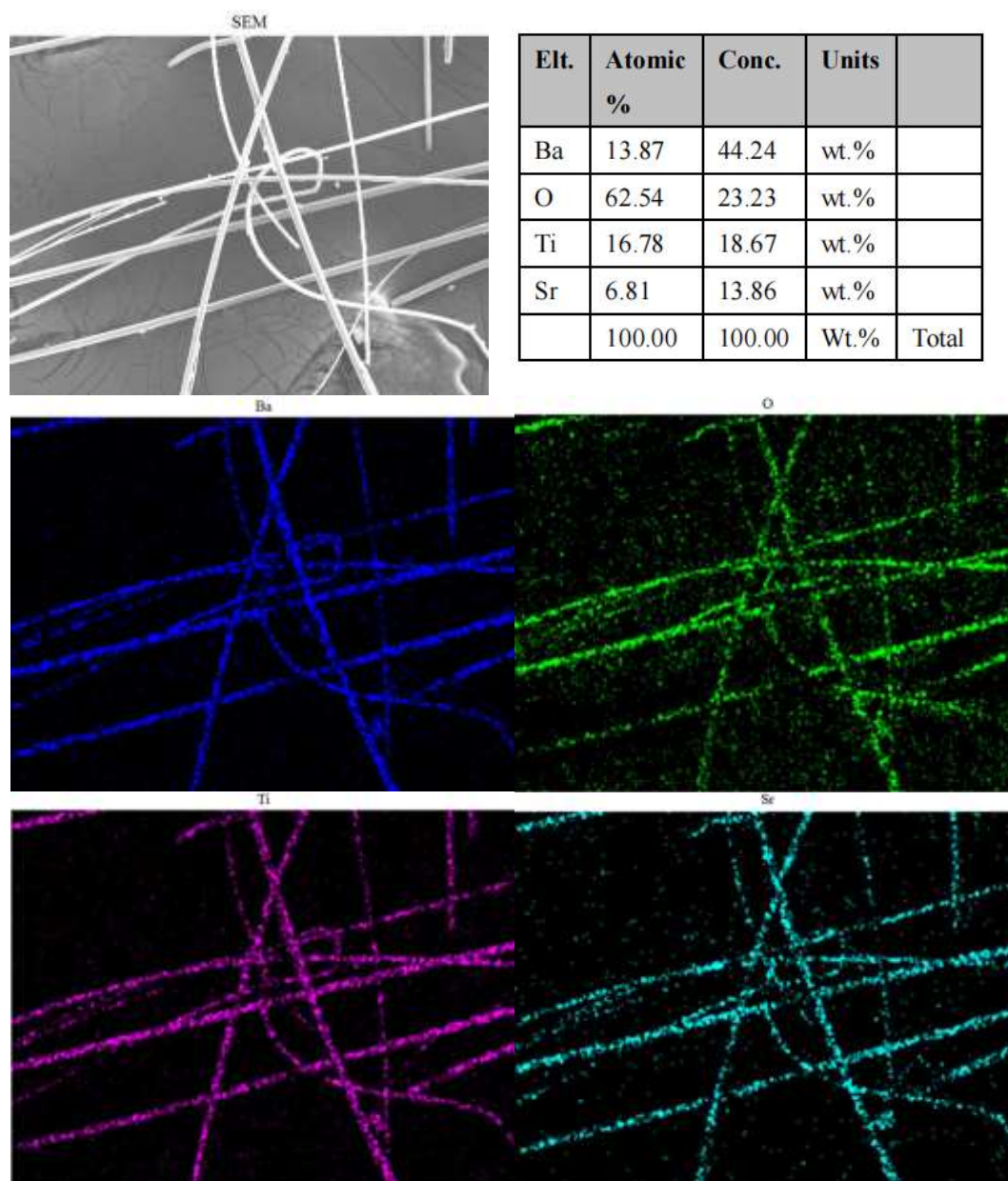


Fig 3: EDS-mapping of fibers

In order to study the synthesis mechanism of $\text{Ba}_{0.7}\text{Sr}_{0.3}\text{TiO}_3$ fibers, particularly the formation mechanism of linear colloids, FI-IR spectra analysis was carried out on the samples prepared in different periods in the NHSG process, and the analysis is shown in Fig. 4(a). $\nu(\text{C-O})$ H bonds disappear in sample (II) $\text{EtOH} + \text{TiCl}_4$ compared with sample (I) EtOH and (II) $\text{EtOH} + \text{TiCl}_4$ [10]. In contrast, Ti (O-C) and Ti-Cl bonds appear at 795 cm^{-1} and 889 cm^{-1} [11]. These certificate the reaction between

anhydrous EtOH and titanium tetrachloride, which is shown as equation (1) [12]. Chlorotitanium ethoxide ($\text{TiCl}_2(\text{OEt})_2$) is the reaction intermediate and has the bimolecular association structure as shown in the left side of Fig. 4 (b) for meeting the hexa-coordination demand of Ti. At the same time, owing to the reaction activities differences of EtO and Cl groups, the CH_3COO prefer to reacts with EtO group for deacetylation reaction. The formation of Ba-O-Ti bond and Sr-O-Ti bond at 1418cm^{-1} [13] and 1340cm^{-1} [14]. Consequently, chlorotitanium ethoxide, $\text{Ba}(\text{OOCCH}_3)_2$ and $\text{Sr}(\text{OOCCH}_3)_2$ ultimately assemble to form linear colloids shown as Fig.2(a) and (b). Fig. 4(b) shows the mechanism for precursors linear-assembling process.

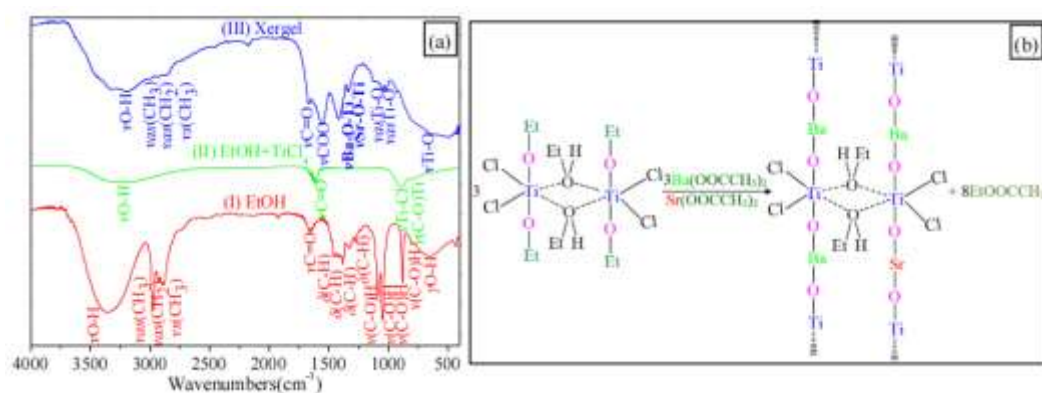


Fig 4: FT-IR spectra of samples (a) and linear-assembly mechanism of precursors (b)

IV. CONCLUSION

Linear self- assembly NHSG method is innovatively applied to prepare novel $\text{Ba}_{0.7}\text{Sr}_{0.3}\text{TiO}_3$ ceramic fibers. The average diameter of $\text{Ba}_{0.7}\text{Sr}_{0.3}\text{TiO}_3$ ceramic fibers are about $2\text{-}2.5\mu\text{m}$, The aspect ratio of $\text{Ba}_{0.7}\text{Sr}_{0.3}\text{TiO}_3$ fibers is more than 10, smooth of surface and solid in cross-section structure with a few spotted defects. Bimolecular association structure of chlorotitanium ethoxide which generated from the reaction of titanium tetrachloride and anhydrous ethanol, which promotes the precursors self-linear-assembly. The precursor self-assembly of non hydrolytic sol-gel method is a new method to prepare ceramic fibers, due to its advantages of excellent molecular economy, high target elements content in the precursor fibers, and not easy to pulverize, fracture and crack in the subsequent ceramic crystallization process.

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